# ZnO(In) SCINTILLATION LIGHT SPECTRA INVESTIGATION FOR HEAVY ION DETECTOR APPLICATION\* $^\dagger$

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### Abstract

ZnO-based ceramics are known as promising scintillators exhibiting light emission in the ultraviolet (UV) spectral region (~390 nm) and ultrafast decay times (<1 ns). They are of great interest for applications in scintillation counters and screens at high-energy heavy ion accelerators. In this contribution, the deterioration of scintillating properties of ZnO-based ceramics subjected to heavy ion exposure at high doses is investigated. The scintillation light spectra of ZnO(In) as a function of fluence for 4.8 MeV/u <sup>48</sup>Ca and <sup>197</sup>Au ions were studied. We observed that the deterioration of the scintillation intensity with increasing fluence follows the Birks-Black model.

## **INTRODUCTION**

Zinc oxide is a multi-functional material with many applications due to its interesting properties [1, 2]. In particular, since the 1960s it has been known as a promising scintillation material that exhibits sub-nanosecond fast light emission at room temperature [3–5]. Previously, pure and doped ZnO has been produced in various forms (powder, thin films, single- and poly-crystals, and ceramics) with the purpose of detecting different types of ionizing radiation [6–9].

Recent studies and technological advances have made it possible to produce bulk pieces of indium- and galliumdoped zinc oxide ceramics (ZnO(In) and ZnO(Ga)) by uniaxial hot pressing in vacuum [10, 11]. These scintillating ceramics become highly interesting for heavy-ion radiation detection at accelerator facilities like GSI and the future FAIR facility in Darmstadt, Germany [12]. ZnO(In) and ZnO(Ga) scintillators are considered as promising candidates to substitute plastic scintillator BC400 commonly used for beam intensity monitoring and spill micro-structure characterization at SIS-18 synchrotron, where ion beams from proton to uranium with energies from 150 MeV/u to 4.5 GeV/u can be obtained.

As a part of a research and development project for a radiation-resistant fast scintillation detector, we report on the performance and in-situ characterization of ZnO(In) ceramics scintillation light spectra change as a function of ion fluence.

# ZnO(In) SCINTILLATION SPECTRA

The investigated ZnO(In) ceramic sampels were produced in the form of  $8 \text{ mm} \times 8 \text{ mm} \times 0.4 \text{ mm}$  size plates at the Joint Stock Company "Research and Production Corporation S.I. Vavilova" (St. Petersburg, Russia). The samples were produced with 0.046 at % In<sup>3+</sup> dopping concentration. Samples density estimated from mass and volume measurements was 4.66 g/cm<sup>2</sup>.

In-situ characterization of scintillation light spectra was performed using 4.8 MeV/u energy <sup>48</sup>Ca and <sup>197</sup>Au ions from UNIversal Linear ACcelerator (UNILAC) of the GSI Helmholtz Center for Heavy Ion Research GmbH (Darmstadt, Germany) [13]. The irradiations were carried out at room temperature in vacuum at a beam incidence angle of 45° with respect to the sample surface.

Scintillation light was collected with a lens placed in front of the ion-incident surface of the sample. The collected light was transfered via a light guide to an Ocean Optics QE-Pro spectrometer. The acquired spectra were corrected for dark counts and normalized by the number of ions that hit the sample during the spectrum acquisition.

Figure 1 shows how the scintillation light spectrum of ZnO(In) ceramic sample changes as a result of 4.8 MeV/u <sup>48</sup>Ca ion irradiation. The spectra have only one emission



Figure 1: Spectra of ZnO(In) ceramic as a function of  $4.8 \text{ MeV/u}^{48}$ Ca ion fluence.

band in the ultraviolet (UV) light region with a maximum at around 387 nm. The observed peak corresponds to the nearband-edge emission (NBE) known for scintillation decay times less than nanosecond. With increasing fluence, the

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intensity of this NBE band drops considerably. No new light emission bands are formed as a result of the ion irradiation, not even at the highest fluences reached,  $5 \times 10^{12}$  Ca-ions/cm<sup>2</sup> and  $2 \times 10^{11}$  Au-ions/cm<sup>2</sup>.

## **BIRKS-BLACK MODEL FIT**

The Birks-Black model [14] was used to fit the observed NBE peak intensity reduction as a function of Ca and Au ion fluence. According to this model, the reduction of the scintillation light intensity can be presented as:

$$\frac{I(\Phi)}{I_0} = \frac{1}{1 + \frac{\Phi}{\Phi_{1/2}}},\tag{1}$$

where  $I_0$  is the initial scintillation light intensity, and  $\Phi_{1/2}$  is the critical fluence at which the scintillation light intensity is reduced by 50% of the initial value.

Figure 2 shows the NBE peak intensity as a function of the Ca and Au ion fluence. The intensity evolution follows the Birks-Black model for both ion species. The critical fluence values extracted from a fit of Eq. (1) to the data are  $3.3 \times 10^{11}$  Ca-ions/cm<sup>2</sup> and  $3.5 \times 10^{9}$  Au-ions/cm<sup>2</sup>, respectively.



Figure 2: Intensity of the NBE peak (387 nm) of the scintillation spectrum as a function of irradiation fluence using <sup>48</sup>Ca ions (full circles) and <sup>197</sup>Au ions (open stars). The dashed line represent fits with Eq. (1) from the Birks-Black model.

### CONCLUSIONS

ZnO-based scintillating ceramics are promising materials to be used for heavy-ion particle detection applications. In this work, ZnO(In) scintillating ceramics were exposed to 4.8 MeV/u <sup>48</sup>Ca and <sup>197</sup>Au ion beams in order to investigate the changes in scintillation light spectrum as a function of swift heavy ion fluence. The scintillation light intensity follows the trend predicted by the Birks-Black model. As a result of high fluence irradiation, no new peak formation was observed up to  $5 \times 10^{12}$  Ca-ions/cm<sup>2</sup> and  $2 \times 10^{11}$  Au-ions/cm<sup>2</sup> irradiation at 4.8 MeV/u energy.

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