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# Status of $\text{MgB}_2$ Coating Studies for SRF Applications

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***16<sup>th</sup> International Conference on RF superconductivity  
Paris, France, 22-27 September 2013***

LA-UR-13-27357

# Acknowledgements

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- $\text{MgB}_2$  samples shown here were provided by STI (Brian Moeckly and Chris Yung) and Temple U. (Xiaoxing Xi and Teng Tan)
- KEK provided copper surrogate cavities.
- JLAB provided a Nb surrogate cavity.
- Cornell and FNAL provided their Nb disks and cavities, respectively, and FNAL offered to test  $\text{MgB}_2$  coated cavities with diagnostics.
- Dan Oates of MIT for the email today!

# Outline

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- Background and motivation to use  $\text{MgB}_2$ .
- Brief summary of the characteristics of  $\text{MgB}_2$  thin films comparing to Nb bulk and films
- Coating system at LANL
- Conclusions

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# Background and motivation

# Background

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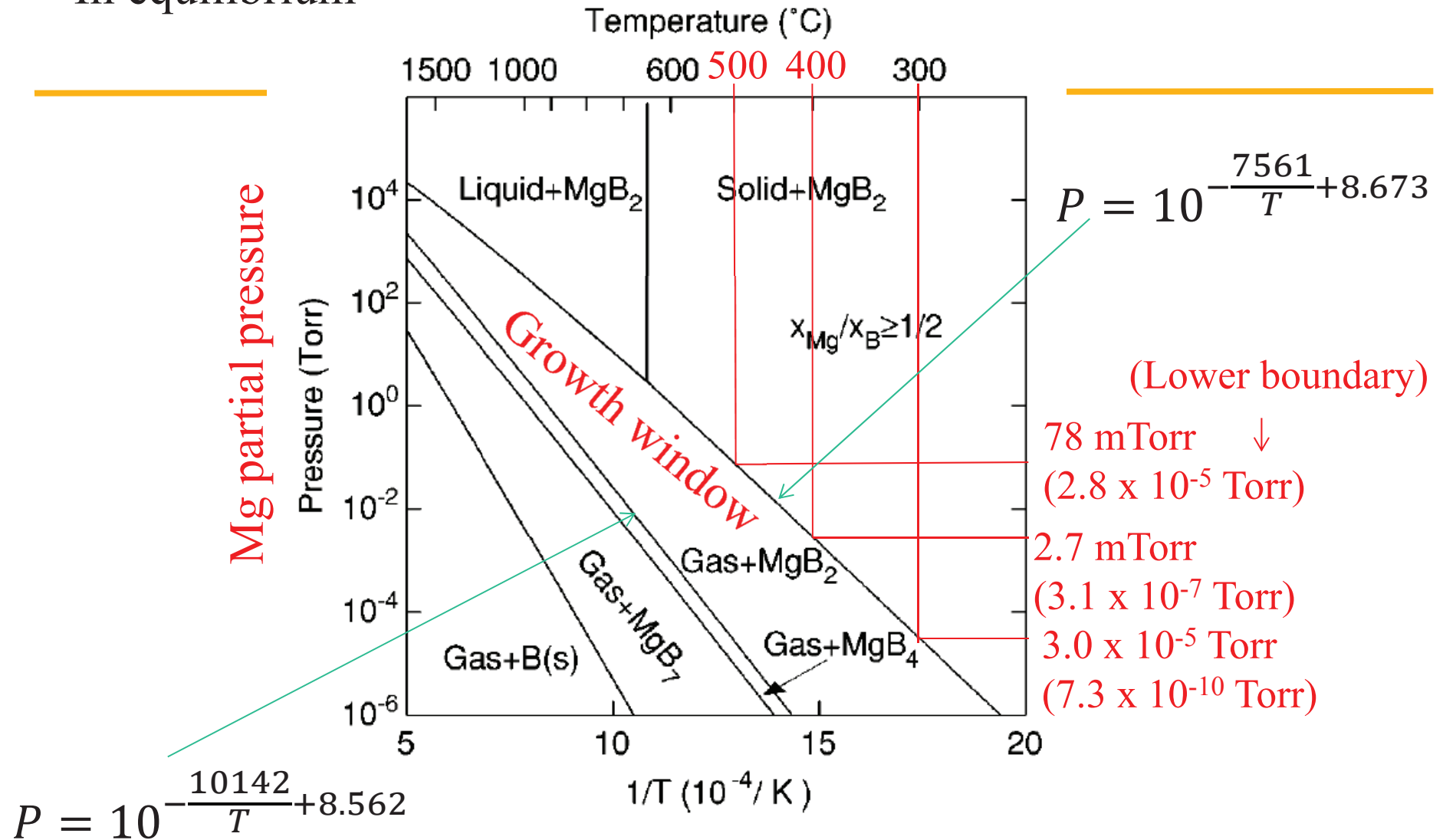
- Nb technology has become close to its full capacity on gradient ( $E_{\text{acc}} \sim 50 \text{ MV/m}$  for electron accelerator) due to its well known critical fields ( $B_{c1} \sim 170 \text{ mT}$  and  $B_c \sim 200 \text{ mT}$ )
- New materials with Gurevich's multi-layer idea could produce cavities that exceed the performance of Nb cavities and open up more opportunities to use SRF cavities.

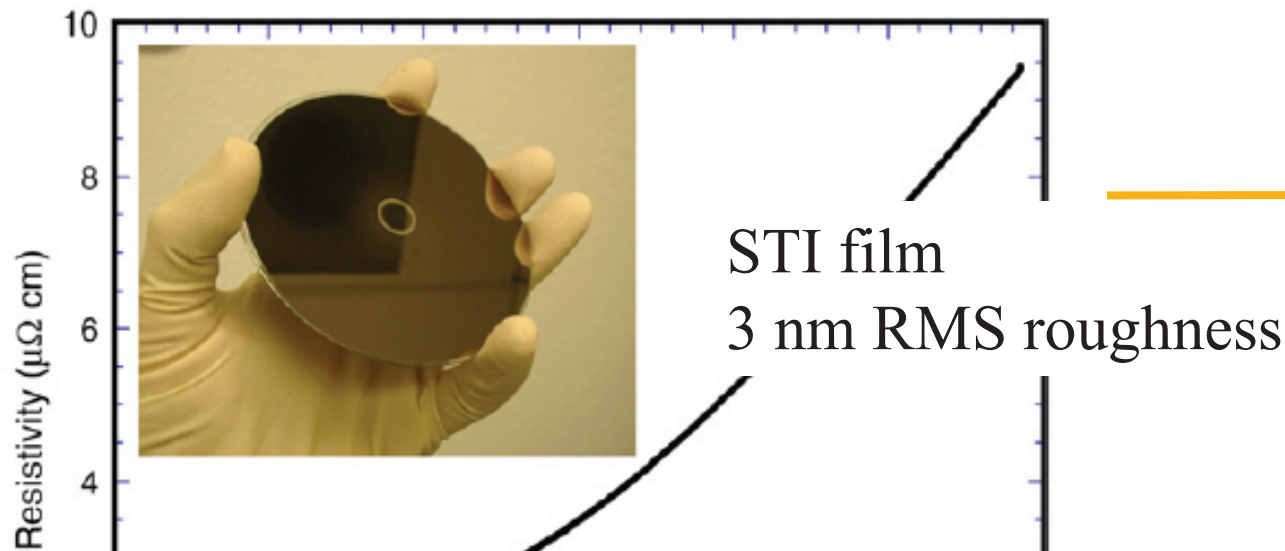
# Motivation to use $\text{MgB}_2$

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- $\text{MgB}_2$  has quite high  $T_c \sim 40$  K compared to other conventional s-wave superconductors ( $< 20$  K), leading to lower BCS resistance and higher thermal tolerance.
- Absence of weak links (low dependence of  $Q_0$  on  $H$ )
- Simple chemical composition
- Can be coated at a large range of temperatures ( $> \sim 250$  °C)

In equilibrium





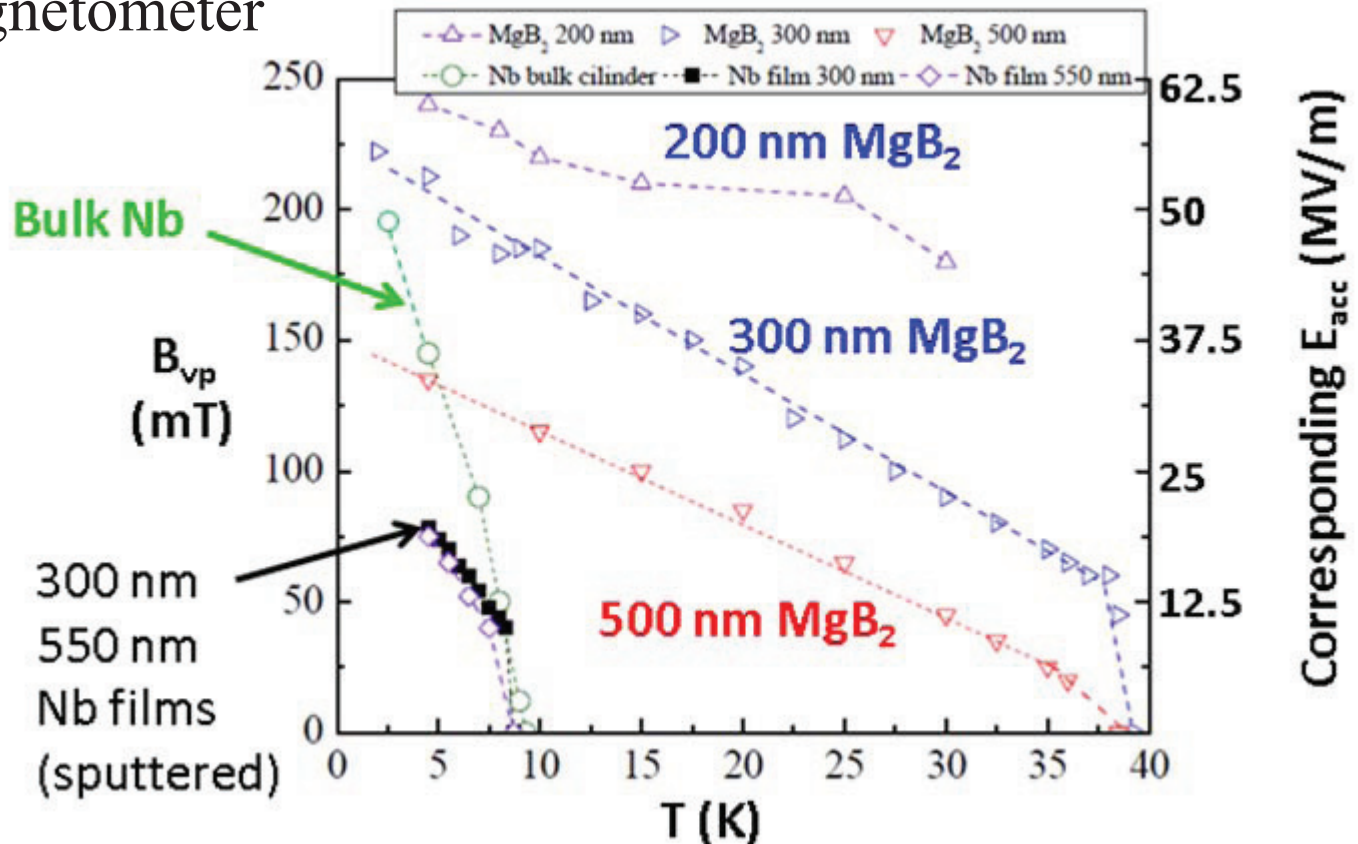
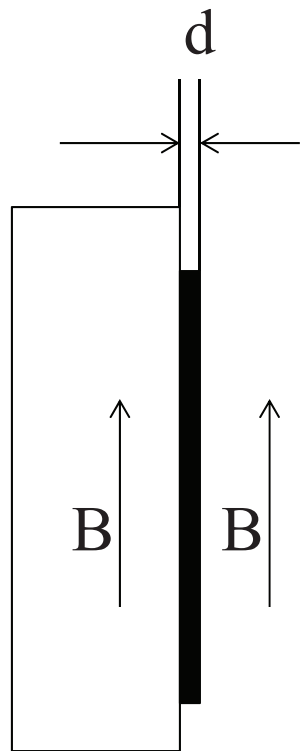
## A brief summary of the characteristics of $\text{MgB}_2$ thin films compared to bulk and film Nb

**Figure 2.** Resistivity versus temperature for an  $\text{MgB}_2$  film deposited on polycrystalline alumina. This film was grown at  $550^\circ\text{C}$  to a thickness of 550 nm and has a zero-resistance  $T_c$  value of 39.1 K. The inset shows a photograph of an  $\text{MgB}_2$  film grown on a 4 inch diameter *r*-plane sapphire substrate.

# Summary of $B_{vp}$ for STI films (200, 300 and 500 nm) compared with cavity-grade bulk Nb and sputtered Nb films

**$MgB_2$  thin films show remarkably high  $B_{vp}$  !!**

SQUID magnetometer

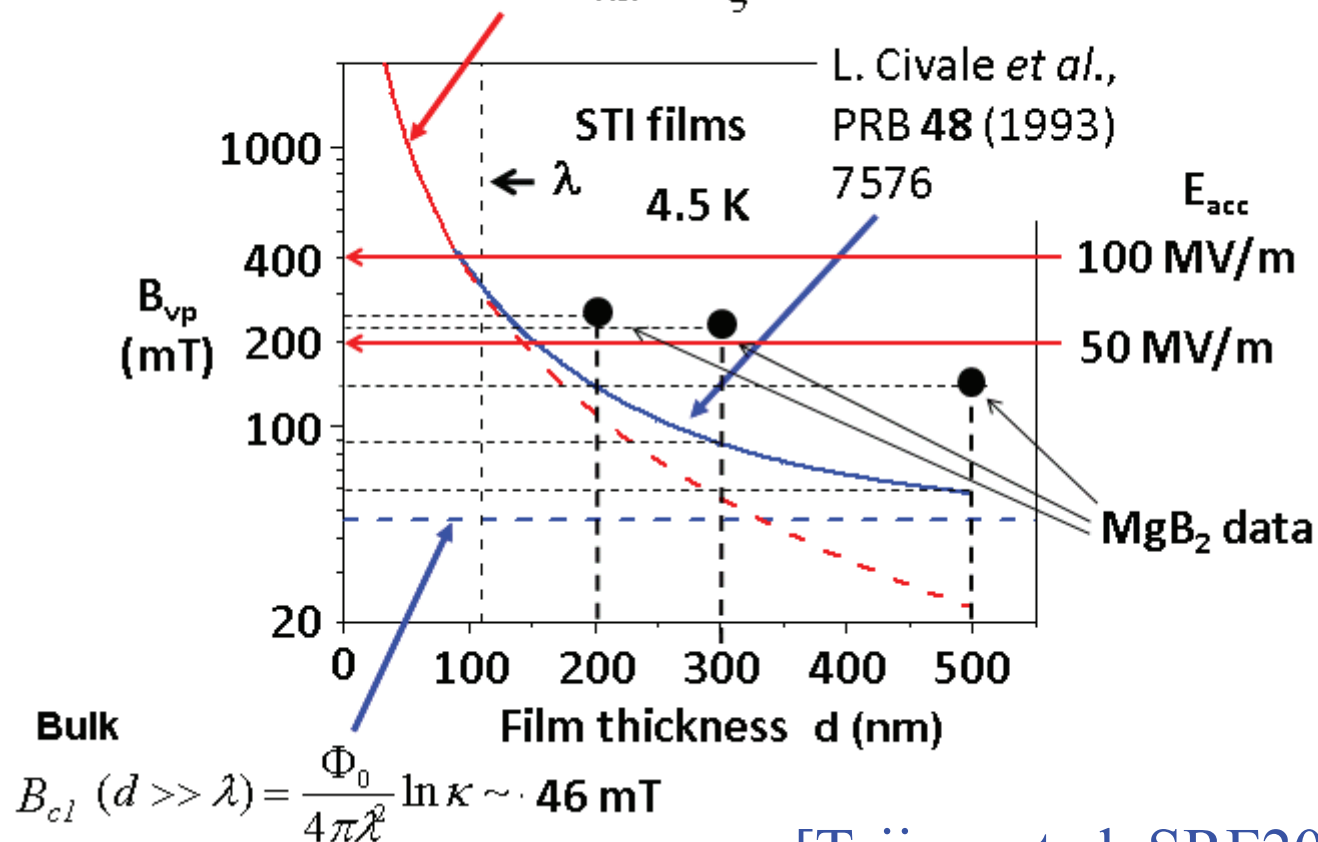


Assume  $B_{peak}/E_{acc} = 4 \text{ mT}/(\text{MV/m})$

[Tajima et al. SRF2011] modified

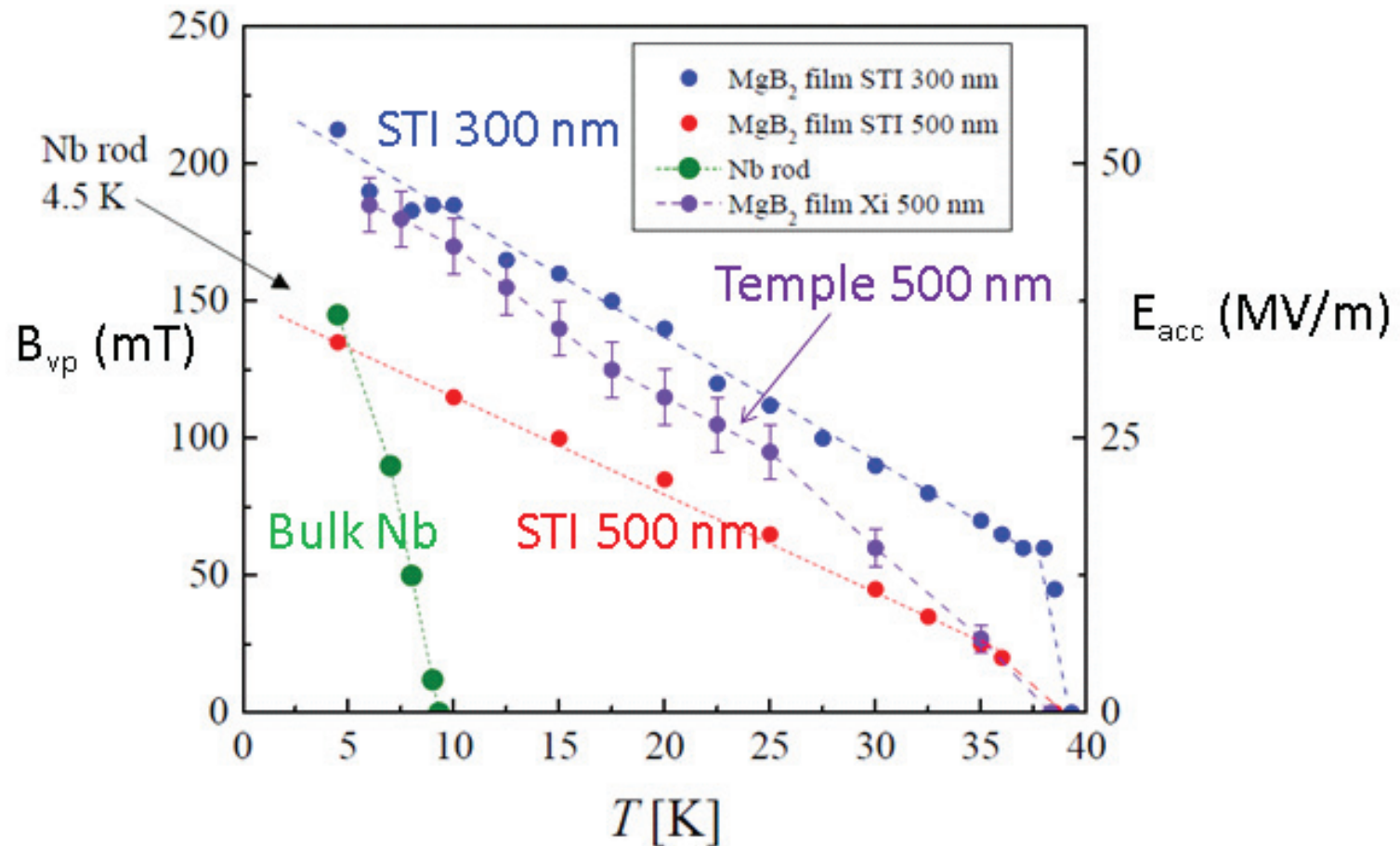
# Comparison with theoretical curve of $B_{c1}$ for thin films assuming $\lambda = 110$ nm and $\xi = 6$ nm.

$$B_{c1} (d \ll \lambda) \approx \frac{2\Phi_0}{\pi d^2} \ln \frac{d}{\xi} \quad \text{Gurevich, APL 88 (2006) 012511}$$

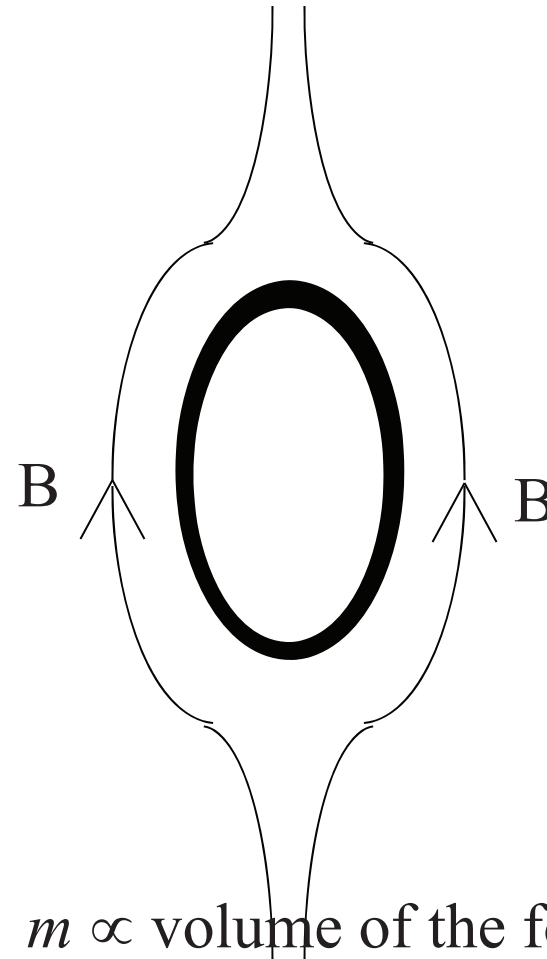
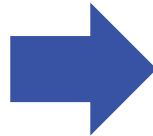
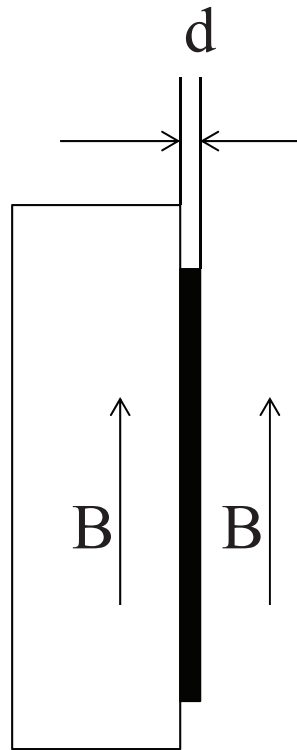


[Tajima et al. SRF2011] modified

# Temple University (Xiaoxing Xi's group) samples prepared with hybrid physical chemical vapor deposition (HPCVD) showed the highest $B_{vp}$ at the same thickness



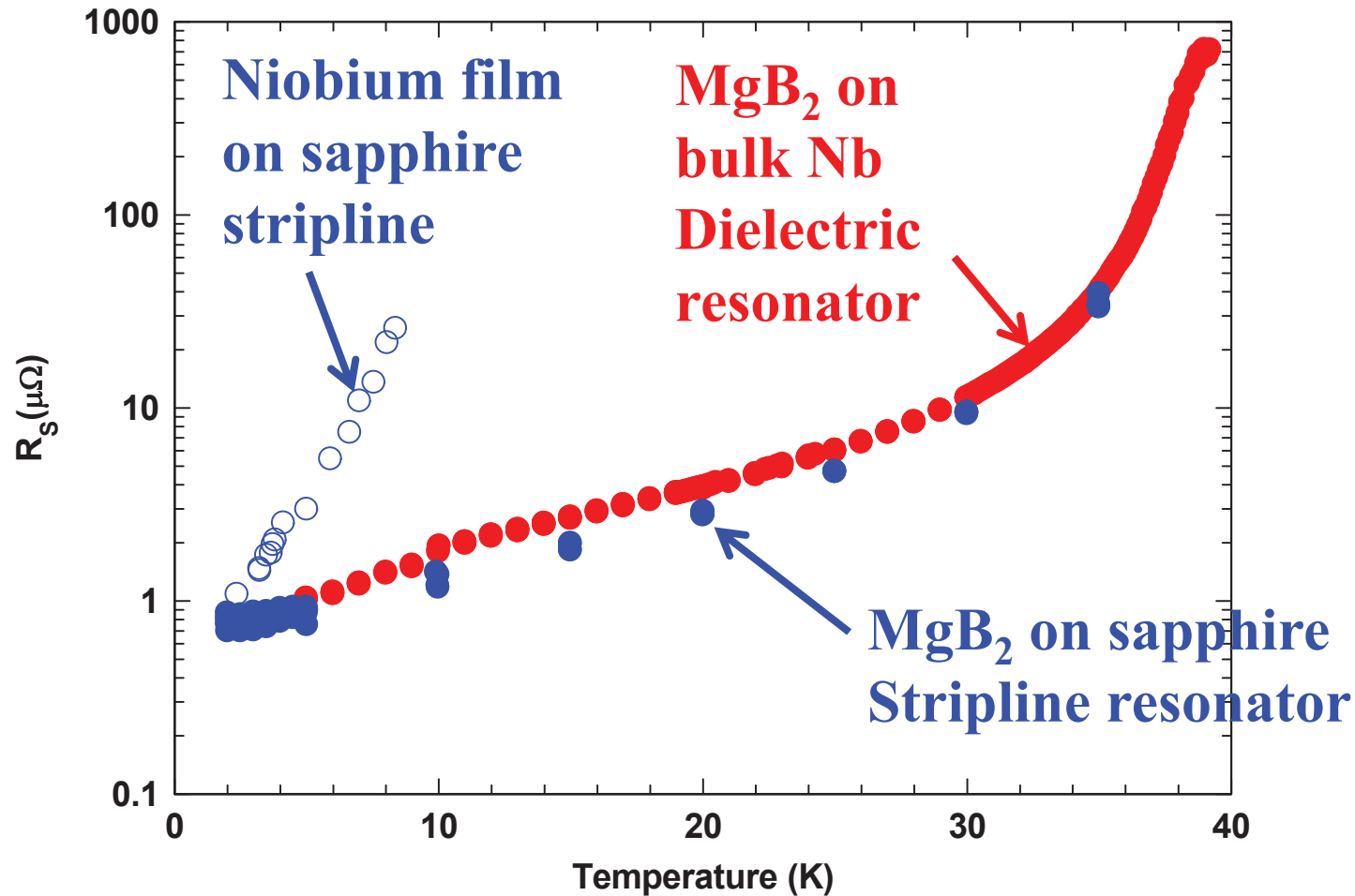
# Our plan to overcome the difficulty of measuring ultra-thin (<200 nm) films and realize cavity-like configuration



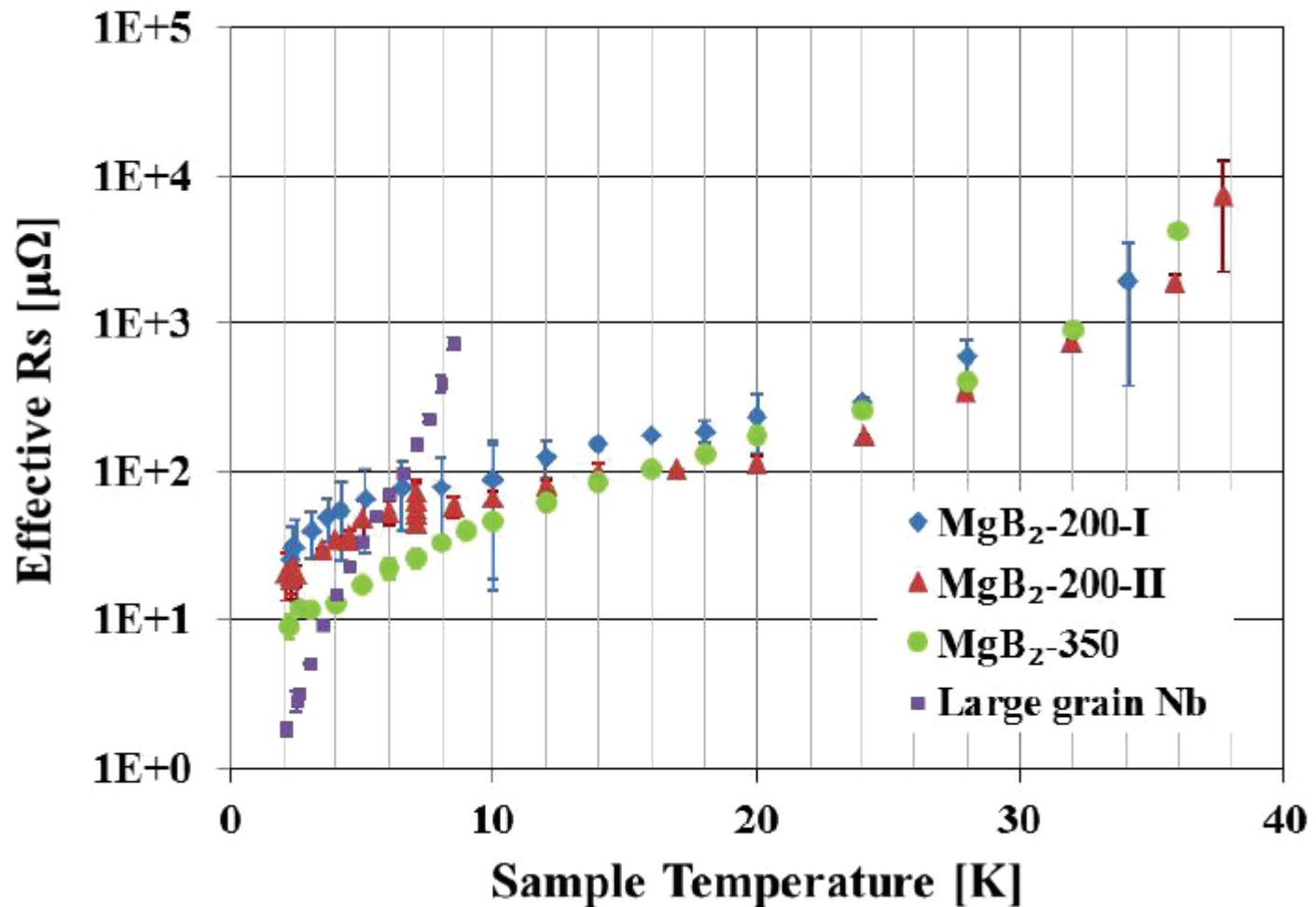
Magnetic moment  $m$   
 $\propto$  volume of film (**very small**)

$m \propto$  volume of the football  
shape (**very large**)

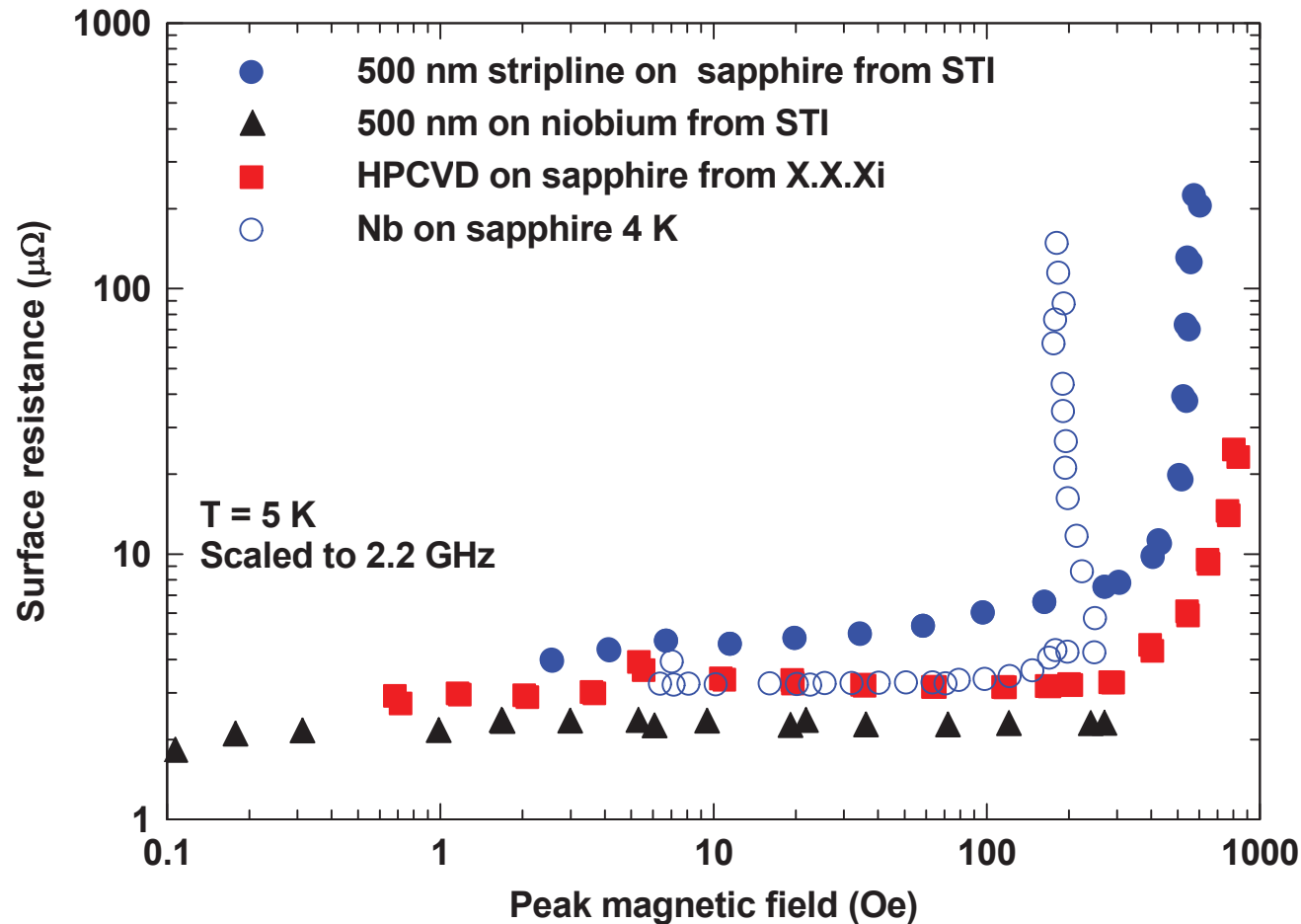
# Low-power $R_s$ (T) test results at MIT



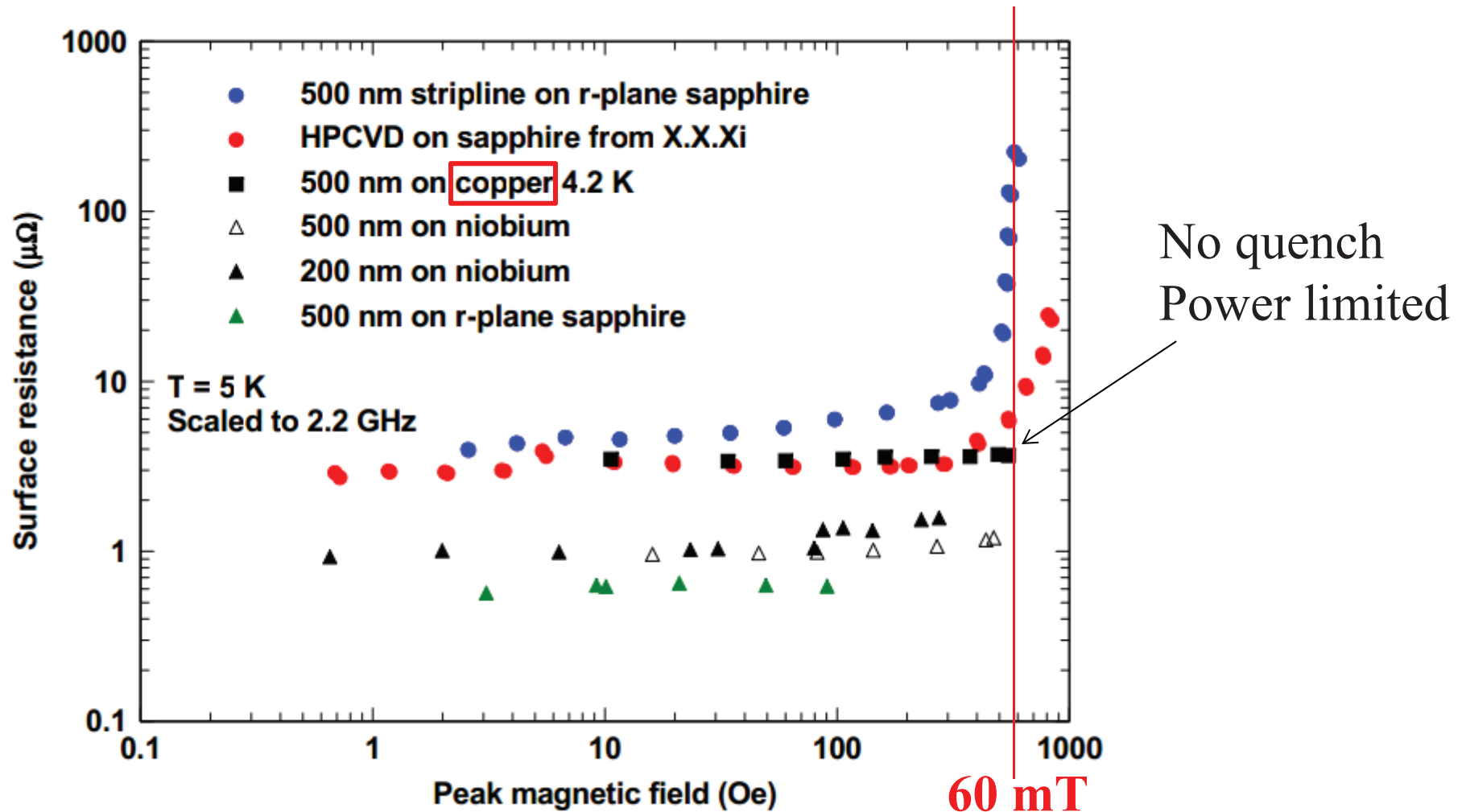
# $R_s(T)$ measured at JLAB



# $R_s$ (H) measured at MIT



# More recent data from MIT



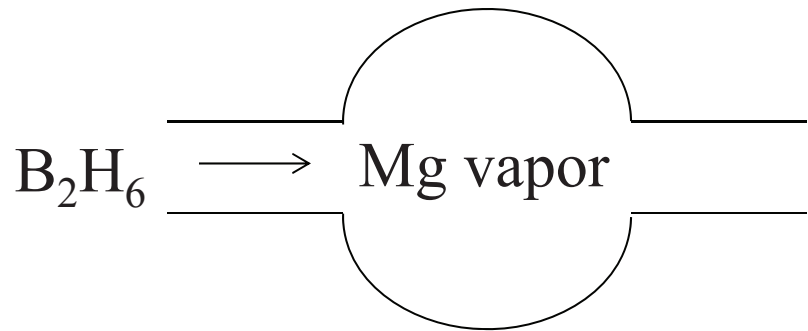
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# Coating system at LANL

# HPCVD was selected because

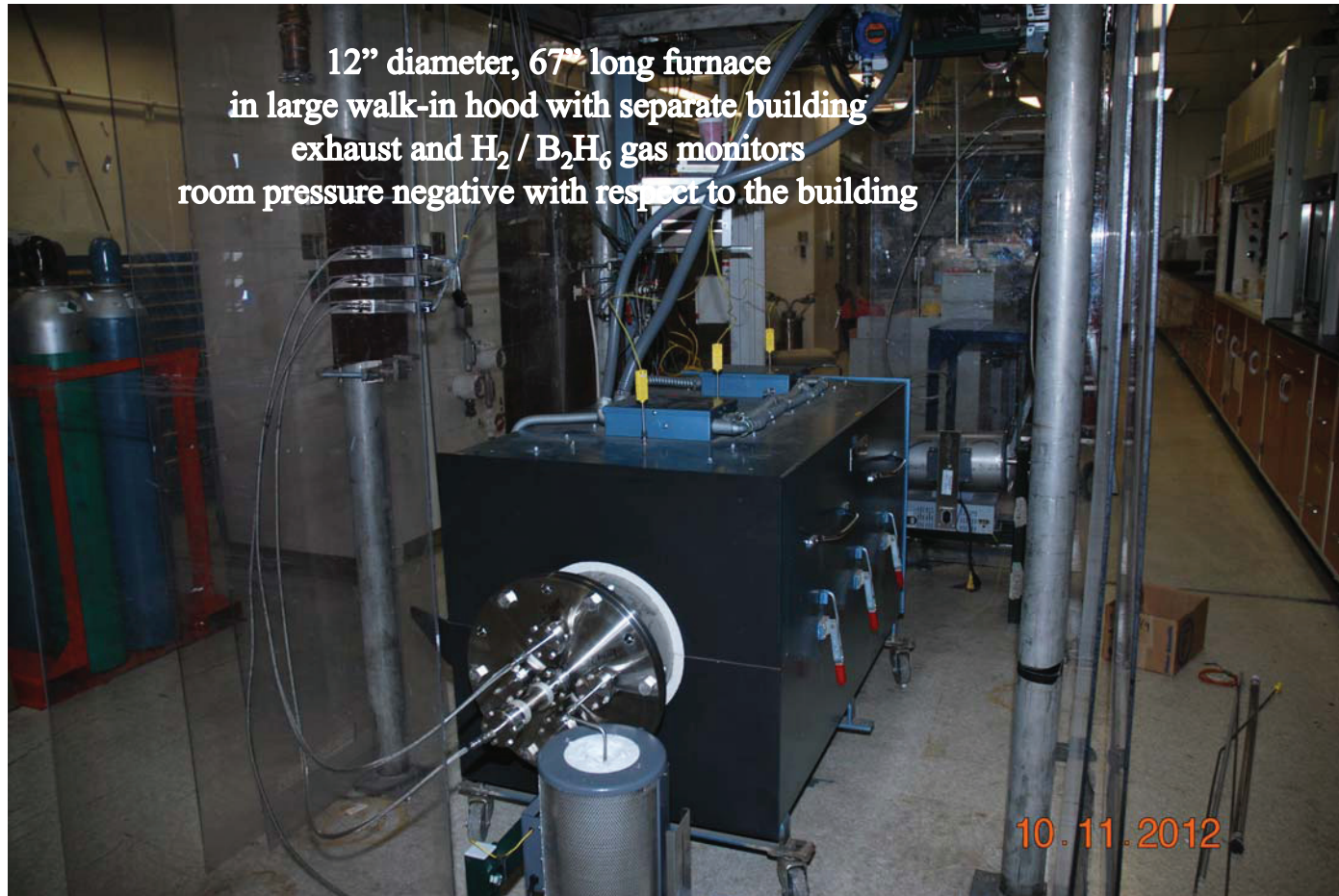
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- Showed highest  $T_c$
- Showed highest  $B_{vp}$
- No UHV is required

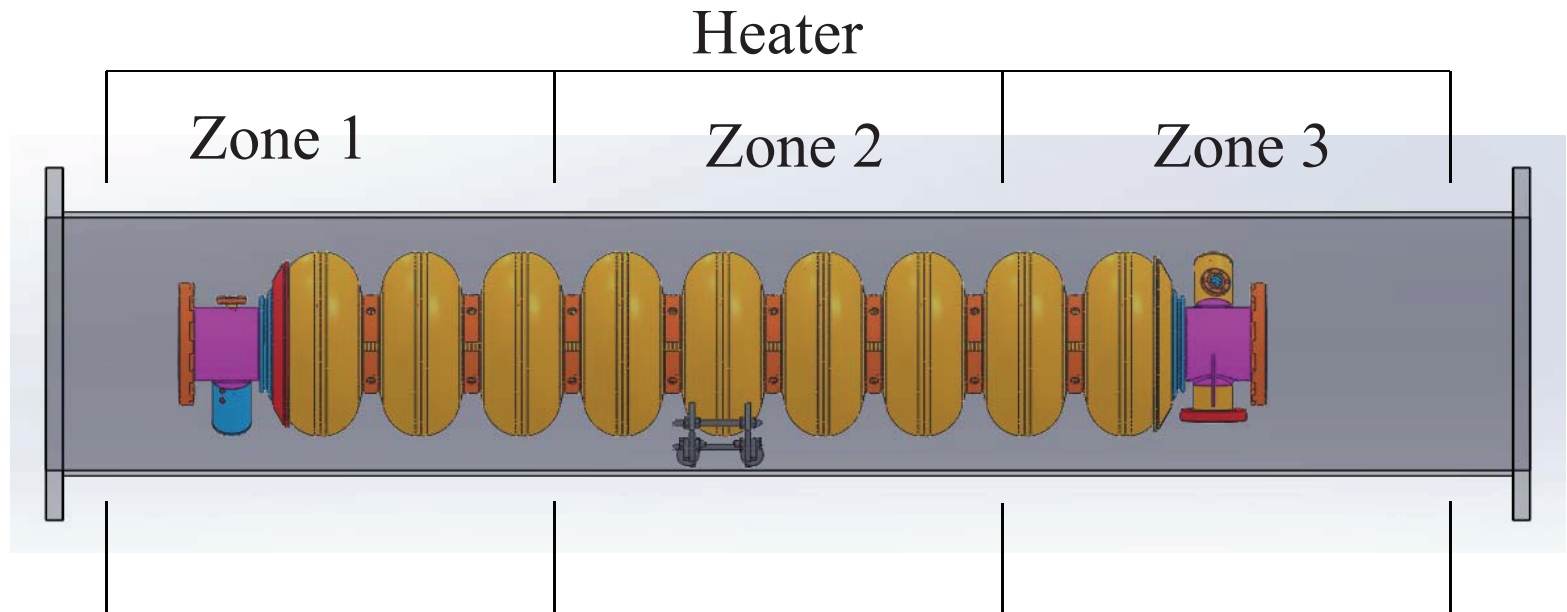


Need to be very careful about diborane ( $B_2H_6$ ) gas due to its toxicity and flammability!

# MgB<sub>2</sub> CVD reactor located at Technical Area 35 at LANL (unfortunately only US citizens allowed)



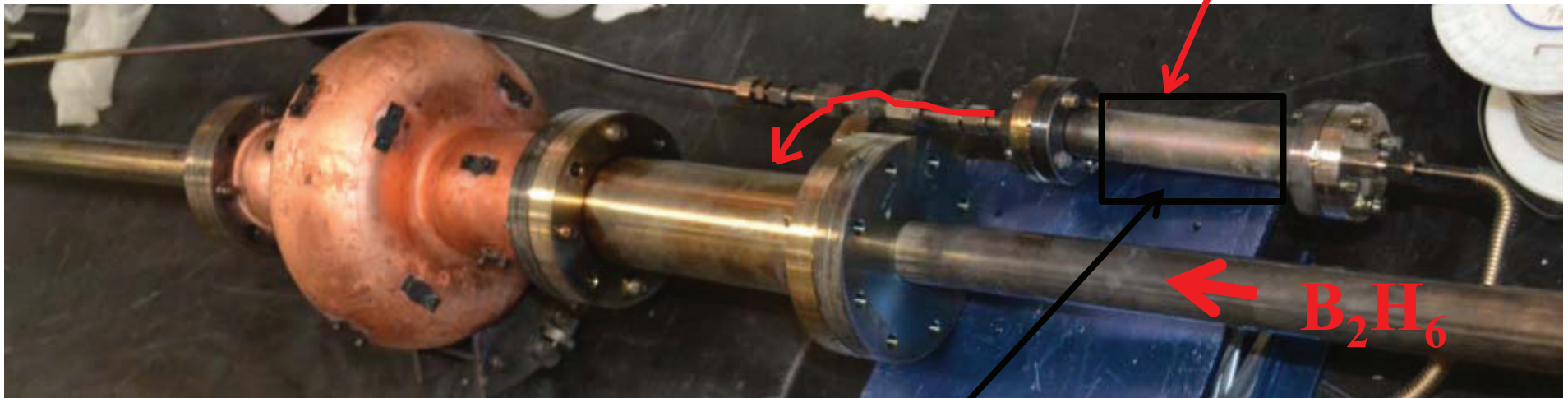
# Furnace is big enough for a 1.3 GHz 9-cell cavity



Currently a 1-cell Cu surrogate cavity is used with 6 mm × 6 mm (sapphire and Nb) samples attached in the holes on the surface

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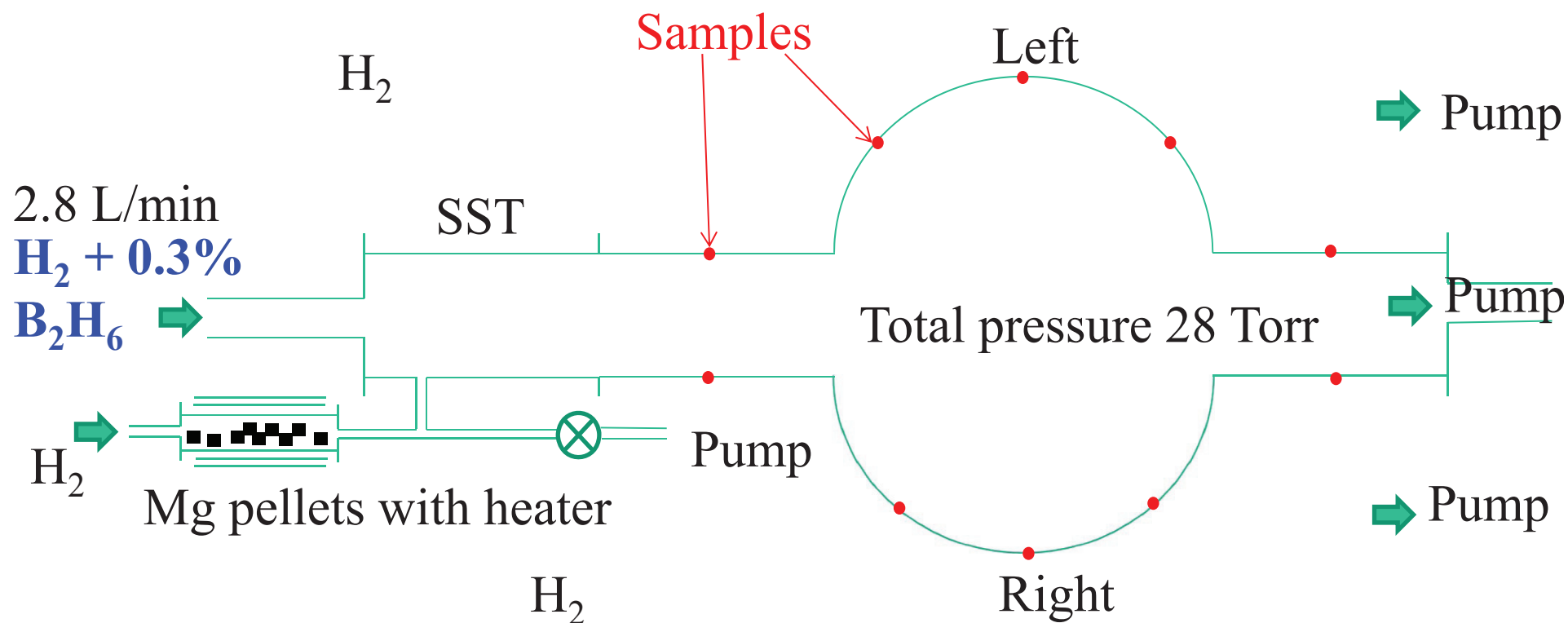
Mg pellets in this



$B_2H_6$

Local heater (not shown)

# Schematic of present coating system (Looking from Top)

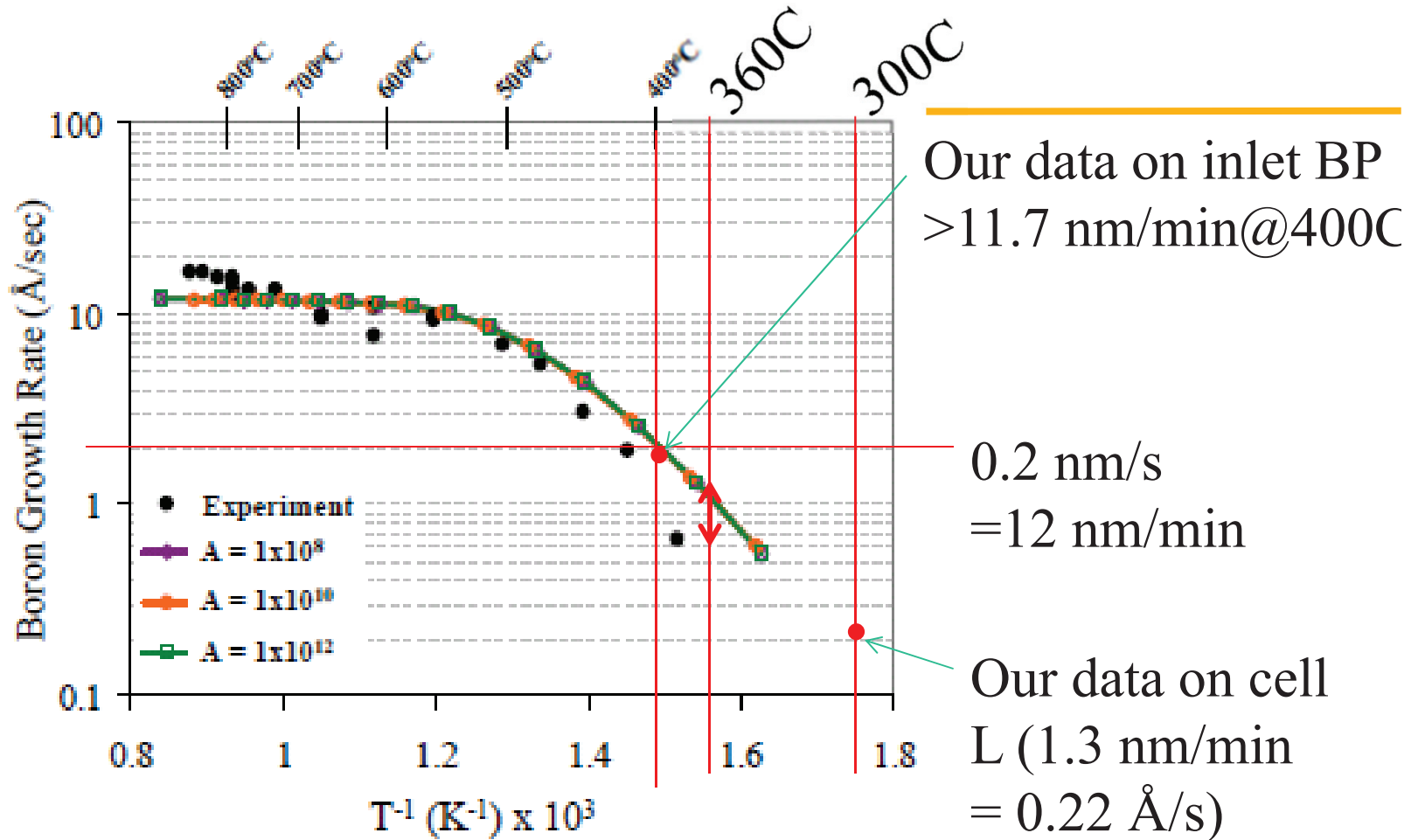


## Learned facts after 10 runs at different furnace T (300-600 °C) and Mg source T (400-840 °C)

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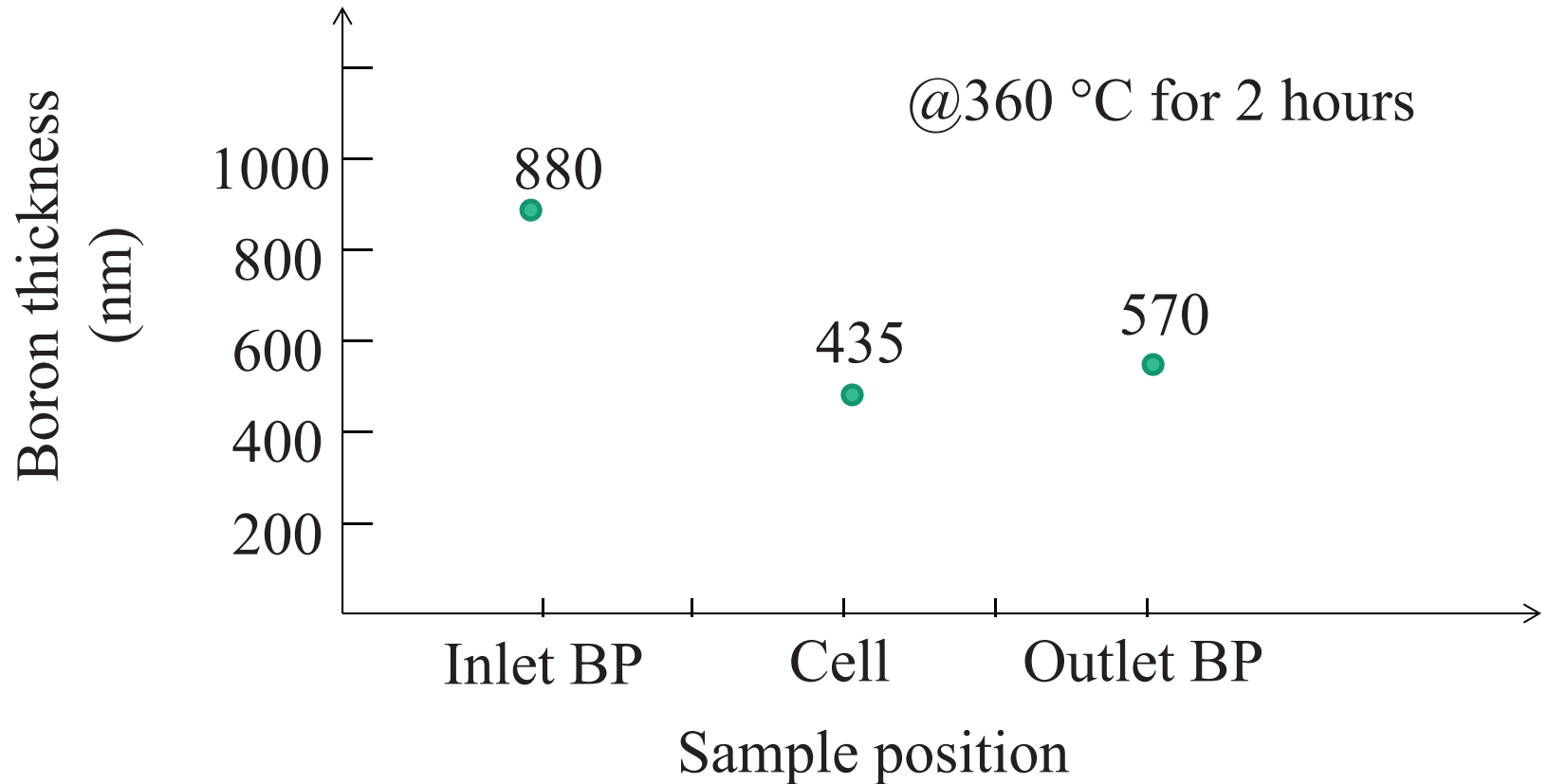
- Boron was not coated on the cavity surface at  $\geq 500$  °C ( $B_2H_6$  gas was likely to be decomposed and consumed on the long inlet line before reaching the cavity)
- Insufficient Mg vapor pressure on the cavity surface. Even heating Mg pellets to  $\sim 840$  °C with a local heater (Mg melting temperature is 650 °C) did not form  $MgB_2$  on the 400 °C cavity surface, while  $B_2H_6$  was flowing.

# Boron growth rate vs. Temperature



[from D.R. Lamborn, PhD thesis, August 2007]

# B thickness profile



# Future options

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- Move the location of Mg source into the SST pipe attached to the cavity so that sufficient Mg vapor reaches cavity surface and maintain the coating temperature at 550 °C or lower.
- 2-stage coating (coat B then react with Mg) at a high temperature ( $\geq 700$  °C, although  $< 800$  °C is preferable to avoid Nb re-crystallization when coated on Nb) (Mg diffused into B only about 20 nm at 600 °C.)

# Conclusions

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- $\text{MgB}_2$  thin film samples especially prepared by HPCVD have shown excellent properties relevant to SRF applications, which warrants the coating of practical-size cavities and study the performance as a cavity.
- LANL as well as other institutes and companies are trying to develop a suitable technique to coat cavities and hope to produce a lot of cavity results by the next SRF conference.